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Chemical state analysis of simulated corium debris by EXAFS

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The chemical state of uranium, zirconium, iron and lanthanide elements in simulated corium debris samples prepared in lab-scale were investigated by synchrotron radiation based extended X-ray absorption fine structure (EXAFS) technique. According to the EXAFS analysis of uranium, their most likely local structure was cubic like fluorite UO_2 . Exceptionally, pentavalent uranium was observed in the $\text{UO}_2\text{-ZrO}_2\text{-Fe}_2\text{O}_3\text{-CaO}$ sample, in which the nearest U-O distance was clearly shorter than those of other samples containing tetravalent uranium. The local structure around zirconium could be classified into cubic, tetragonal and calcia stabilized zirconia with CaO. The oxidation state of iron was divalent for the most samples and a metallic phase was also detected in some of them. The small amount lanthanide elements (Nd and Gd) in the simulated debris seem to exist as a solid solution with UO_2 .

Keywords: simulated corium debris; EXAFS; synchrotron radiation

1. Introduction

The dominant chemical form of uranium in the solidified fuel debris is thought to be $(\text{U,Zr})\text{O}_{2\pm x}$ solid solution in the case of Fukushima Daiichi NPS accident [1,2]. Iron, calcium, and lanthanides derived from the structural materials and fission products are expected to be included as minor soluble components. It is important to clarify the chemical state of each element contained in the fuel debris in order to determine methods of retrieval operation and storage. The EXAFS technique is one of the useful tools since information on each element can be selectively obtained from multicomponent mixtures such as the debris. In the present work, the chemical state of uranium, zirconium, iron and lanthanide elements (Nd and Gd) in the simulated corium debris samples prepared in lab-scale was evaluated from synchrotron radiation based EXAFS analysis.

2. Experimental

2.1. Simulated corium debris samples

Twelve simulated corium debris samples containing uranium was prepared in the present work. They are oxide solid solution of uranium, zirconium, iron, gadolinium, neodymium and calcium. The simulated debris samples used in the present work are summarized in **Figure 1**. The samples UO_2 , U_3O_8 and three kinds of $\text{UO}_2\text{-ZrO}_2$ solid solution (A1-A5) can be regarded as

fundamental materials. In addition, Fe and Ca were added assuming the reaction with reactor pressure vessel and concrete material. As fission product component, Gd and Nd were used in the present work. The samples adding Fe, Zr and Ca to UO_2 are marked as B1-B5. The samples adding Gd and Nd to UO_2 , B3 and B4 and some related samples are defined as C1-C7. The samples were prepared by sintering of the mixture pellet of each oxide powder (UO_2 , ZrO_2 , Fe_2O_3 , CaO, Gd_2O_3 and Nd_2O_3) at 1873-1973K for 6 hours in Ar gas (10-30ppm purity) flow. The powder sample taken from the pellet was used in the EXAFS measurement. Three samples; monoclinic and tetragonal ZrO_2 and calcia stabilized zirconia $\text{Zr}_{0.84}\text{Ca}_{0.16}\text{O}_{1.84}$ were used as standard materials.

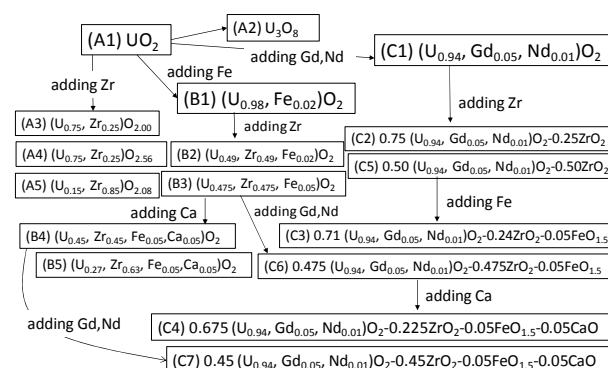


Figure 1. Tree diagram of the simulated corium debris samples used in the EXAFS analysis.

2.2. EXAFS analysis

The XAFS measurement was performed at the

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BL-27B station of KEK, Tsukuba, Japan. X-ray beam monochromatized by Si(111) double crystals is available in the beamline. The XAFS data of the U L₃-edge, Zr K-edge, Fe K-edge, Gd L₃-edge, Nd L₃-edge was obtained using transmission and fluorescence mode. The XAFS data was analyzed by WinXAS Ver.3.2[3] in order to obtain EXAFS function $k^3\chi(k)$ and Fourier transform magnitude |FT|. The structural parameter like coordination number and interatomic distance was obtained by the curve fitting procedure in the WinXAS. The correction parameter required in the fitting analysis; phase shift and backscattering amplitude was obtained from the XAFS simulation software FEFF Ver.8.4 [4].

In the present work, we have to pay attention to the limitation in the EXAFS analysis of multiphase sample. According to the phase diagram of UO₂-ZrO₂ system [5,6], the sample are unlikely to be single phase solid solution. In most cases, UO₂-ZrO₂ solid solutions consist of at least two phases; typically uranium rich and zirconium rich phases. Then it is natural to consider that the EXAFS of uranium reflects signals from uranium rich phase of the sample.

3. Results and discussion

3.1. Adding Zr to UO₂ and O/M dependence

The U L₃-edge EXAFS functions and their FT magnitudes of the samples A3-A5 are shown in **Figure 2**, together with those of UO₂(A1) and U₃O₈(A2). The EXAFS results of the A3 sample (U_{0.75}Zr_{0.25})O_{2.00} are close to those of the UO₂ fluorite structure, though the amplitude in k-space and the peak height in R-space are slightly lower than those of the A1 sample UO₂. It can be considered that the cubic structure like UO₂ was kept in the A3 sample. The A4 sample (U_{0.75}Zr_{0.25})O_{2.56} and the A5 sample (U_{0.15}Zr_{0.85})O_{2.08} resulted in slightly shorter U-O distance in the FT magnitude as shown in the Figure 1. The fitting values 2.18 Å for the A4 sample and 2.22 Å for the A5 sample are clearly shorter than 2.34 Å for the A1 sample (UO₂) but are close to 2.22 Å for the A2 sample (U₃O₈). It means that uranium in the A4 and A5 samples is not tetravalent. The 2nd peak position of the A4 sample is close to that of U₃O₈. On the other hand, the 2nd peak of the A5 sample was very weak and its position was shorter than that of U₃O₈.

The Zr K-edge EXAFS functions and their FT magnitudes of the samples A3-A5 are shown in **Figure 3**, together with those of tetragonal and monoclinic ZrO₂ and calcia stabilized zirconia (CSZ). The EXAFS function of the A3 sample is not in agreement with that of monoclinic ZrO₂ [7] as can be seen in the Figure 3. According to the curve fitting analysis of the 1st peak, the nearest Zr-O distance was 2.18 Å for the A3 sample. This value is slightly longer than those of tetragonal ZrO₂ (2.065 Å) [8] and CSZ (2.12 Å) [9]. Instead it is close to 2.20 Å for unstable cubic ZrO₂ [7]. The 2nd peak of the A3 sample was not singlet but split into three correlations. We can easily notice that it is relatively

close to the corresponding function of the U L₃-edge FT data shown in the Figure 2. Therefore it suggests that the local structure around Zr in the A3 sample is cubic. The EXAFS results of the A4 sample are in good agreement with those of CSZ. The EXAFS function of the A5 sample showed very small oscillation. The small oscillation in the EXAFS function may be assigned to disordered structure, multiphase effect or the self absorption effect. Therefore, it is difficult to determine the local structure, though the FT magnitude showed the peak profile similar to CSZ.

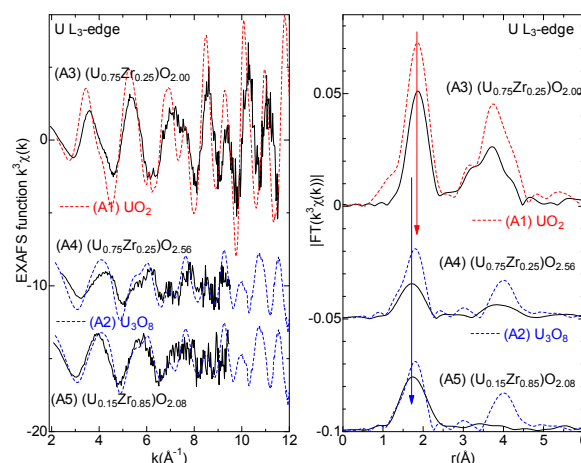


Figure 2. U L₃-edge EXAFS functions and FT magnitude functions of the A1-A5 samples.

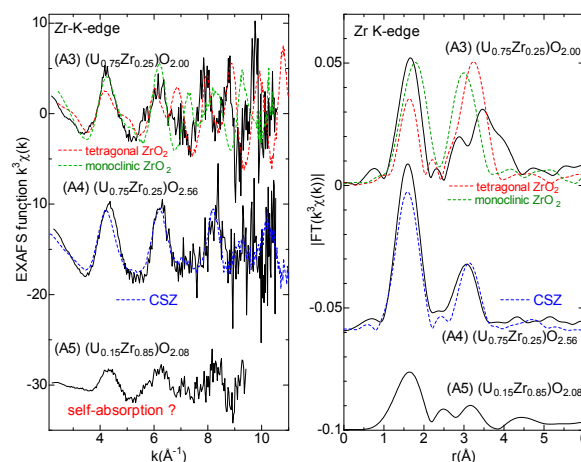


Figure 3. Zr K-edge EXAFS functions and FT magnitude functions of the A3-A5 samples.

3.2. Adding Fe, Zr and Ca to UO₂

In the next step, we made an investigation into the effect of additional elements (Zr and Ca). The U L₃-edge EXAFS functions and their FT magnitudes of the samples B1-B5 are shown in **Figure 4**, together with those of UO₂(A1) and the sample A3. In all the samples, the EXAFS results are similar to those of the sample A3 (U_{0.75}Zr_{0.25})O_{2.00}. It means that the local structure of them is basically cubic. The nearest U-O distance obtained from the curve fitting analysis ranged from 2.32 to 2.35 Å for the B1, B2, B3 and B4 samples.

Only the B5 sample showed different U-O distance, 2.27 Å. It is shorter than that of tetravalent UO_2 but longer than hexavalent UO_3 [10]. It suggests that the oxidation state of uranium is pentavalent, since single phase state was confirmed for the sample B5 from SEM/EDX and X-ray diffraction analyses.

The Zr K-edge EXAFS functions and their FT magnitudes of the samples B2-B5 are shown in **Figure 5**. The obtained EXAFS results are in good agreement with either of the two results of the A3 sample and CSZ. It can be clear that the local structure around zirconium is tetragonal with iron oxide and CSZ with calcium.

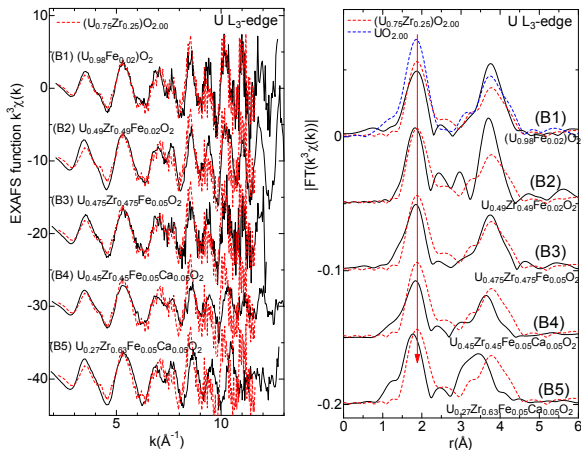


Figure 4. U L_3 -edge EXAFS functions and FT magnitude functions of the B1-B5 samples.

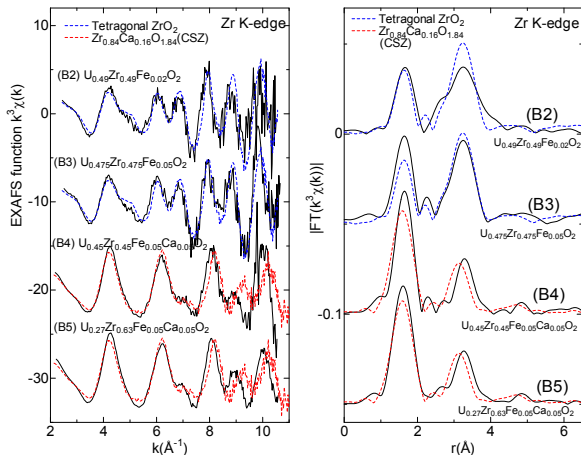


Figure 5. Zr K-edge EXAFS functions and FT magnitude functions of the B2-B5 samples.

The normalized Fe K-edge XANES spectra and the FT magnitudes of the samples B1-B5 are shown in **Figure 6**, together with those of iron standard samples. In comparison with spectra of divalent FeO and trivalent Fe_2O_3 , the oxidation state of iron in all the samples is thought to be mainly divalent. The 1st peak can be assigned to the Fe-O of divalent FeO from the curve fitting analysis. The 2nd peak which is observed near the 1st peak in the FT magnitudes of the sample B1 and B2 suggests that these samples are a mixture with iron metal, since its distance (2.45 Å) is shorter than Fe-Fe

separation (3.05 Å) in FeO .

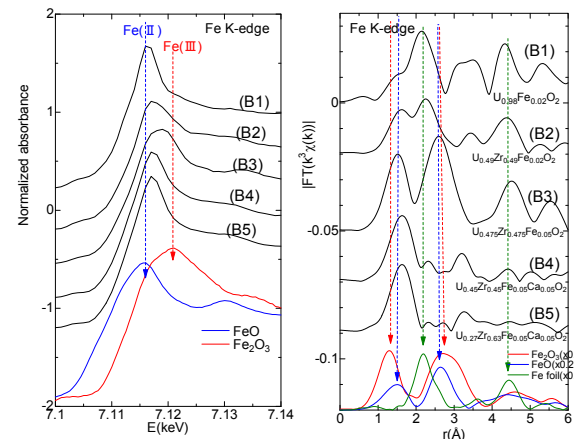


Figure 6. Fe K-edge normalized XANES spectra and FT magnitude functions of the B1-B5 samples.

3.3. Adding lanthanides, Zr, Fe and Ca to UO_2

The U L_3 -edge and Zr K-edge EXAFS functions of the samples C1-C7 samples are shown in **Figure 7**. The function of the B1 sample is in good agreement with that of fluorite UO_2 . It is considered that the cubic structure observed in the sample A3 ($\text{U}_{0.75}\text{Zr}_{0.25}\text{O}_{2.00}$) is predominant in all other samples. The Zr K-edge EXAFS functions in the Figure 7 could be classified into three types. The samples C2 and C5, in which UO_2 , ZrO_2 and lanthanide oxides are contained, resulted in the cubic structure like the sample A3. They changed to tetragonal structure by adding iron oxide as seen in the samples C3 and C6. By adding calcium, CSZ structure was observed in the samples C4 and C7. These results are compatible with conclusion in the preceding section.

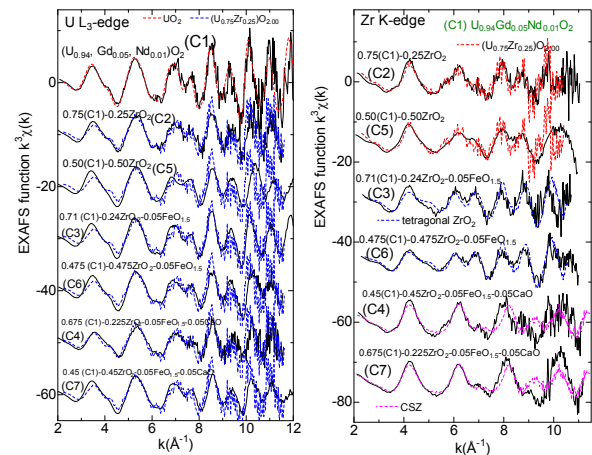


Figure 7. U L_3 -edge and Zr K-edge EXAFS functions of the C1-C7 samples.

The Gd L_3 -edge and Nd L_3 -edge EXAFS functions of the sample C1-C7 and the FT magnitudes of the C1 sample are shown in **Figure 8**. A significant change was not observed in the Gd L_3 -edge EXAFS functions and all were almost the same. It is not in agreement with that of pure Gd_2O_3 . Similar results were obtained in the Nd L_3 -edge results. It is well known that gadolinium makes

solid solution with UO_2 [11]. The FT magnitudes of the sample C1 is relatively close to that of $\text{UO}_2(\text{A1})$. As the results of curve fitting analysis, the 1st Gd-O and Nd-O distances were 2.41 Å and 2.43 Å, respectively. Their coordination number was close to 8. It can be suggested that these lanthanide elements exists as solid solution with UO_2 in the samples C1-C7.

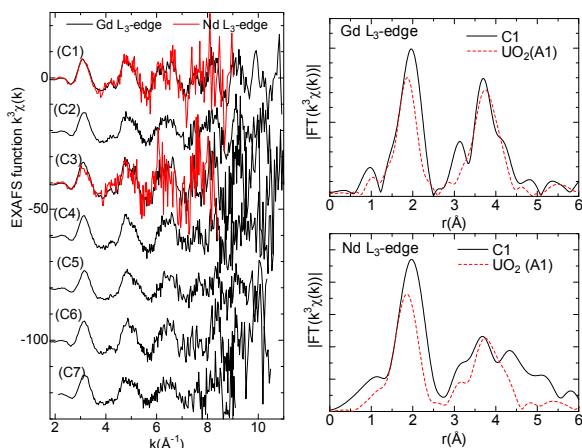


Figure 8. Gd $L_{3\text{-edge}}$ and Nd $L_{3\text{-edge}}$ EXAFS functions of the C1-C7 samples and FT magnitudes of the C1 sample.

4. Conclusion

The chemical state of uranium, zirconium, iron and lanthanides in the simulated corium debris samples were evaluated from the EXAFS analysis (comparison with standard materials and the curve fitting analysis). The results listed in **Table 1** are summarized as follows,

Table 1. Chemical state of uranium, zirconium and iron in the simulated corium debris samples evaluated by the EXAFS analysis (×; not contained, ○:contained).

| | U | Zr | Fe | Ca |
|----|-------------------------|------------|---------------|----|
| A3 | cubic(IV) | cubic | × | × |
| A4 | U_3O_8 | CSZ | × | × |
| A5 | $\text{U}_3\text{O}_8?$ | unsolved | × | × |
| B1 | cubic(IV) | × | metal, Fe(II) | × |
| B2 | cubic(IV) | tetragonal | metal, Fe(II) | × |
| B3 | cubic(IV) | tetragonal | Fe(II) | × |
| B4 | cubic(IV) | CSZ | Fe(II) | ○ |
| B5 | cubic(V) | CSZ | Fe(II) | ○ |
| C1 | fluorite(IV) | × | × | × |
| C2 | cubic(IV) | cubic | × | × |
| C3 | cubic(IV) | tetragonal | Fe(II) | × |
| C4 | cubic(IV) | CSZ | Fe(II) | ○ |
| C5 | cubic(IV) | cubic | × | × |
| C6 | cubic(IV) | tetragonal | Fe(II) | × |
| C7 | cubic(IV) | CSZ | Fe(II) | ○ |

(1)The chemical state of uranium was mainly tetravalent with the cubic structure.

(2)In the sample B5 ($\text{U}_{0.25}\text{Zr}_{0.63}\text{Fe}_{0.05}\text{Ca}_{0.05}\text{O}_2$), the oxidation state of uranium was pentavalent.

(3)The oxidation state of iron was mainly divalent in the most samples. Metallic component was also observed in

the samples B1 and B2.

(4)The chemical form of zirconium was classified into the cubic equivalent to uranium, tetragonal and CSZ. The tetragonal form was found in the samples containing iron and CSZ was observed in the samples with calcium (except for the A4 sample in which calcium is not contained).

(5)The EXAFS results of Gd and Nd in the simulated debris samples suggests cubic structure similar to UO_2 . It can be concluded that chemical state of uranium does not change drastically by the mixing, though pentavalent uranium was observed. On the other hand, the local structure around zirconium changes by mixing with iron and calcium.

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